



# Oxygen plasma treatment WO<sub>3</sub> nanorods for Improvement H<sub>2</sub>S gas sensing

Chaiyan Oros<sup>a,\*</sup>, Phusit Sangpradub<sup>b</sup>, Preeyanut Daunglaor<sup>b</sup>, Thita Yodsawad<sup>b</sup>

<sup>a</sup>Faculty of Science and Technology, Rajamangala University of Technology Rattanakosin, Nakornpathom, 73170 Thailand

<sup>b</sup>Faculty of Liberal-Arts, Rajamangala University of Technology Rattanakosin, Nakornpathom, 73170 Thailand

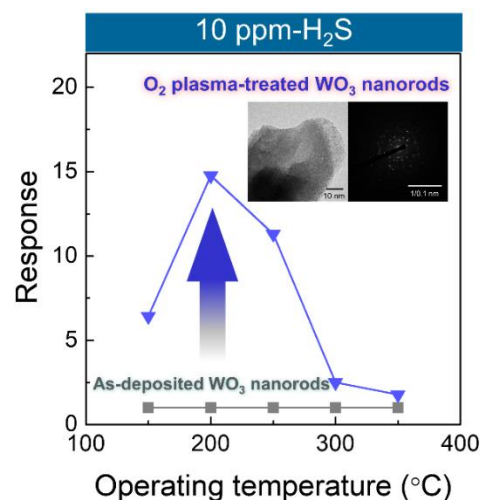
\*Corresponding Author: chaiyan.oro@rmutr.ac.th

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## Abstract

Herein, the oxygen (O<sub>2</sub>) plasma has been used to post-treat tungsten oxide (WO<sub>3</sub>) nanorods to improve the sensing performance of the H<sub>2</sub>S gas sensor. The reactive DC magnetron sputtering process with the glancing-angle deposition (GLAD) technique was used to prepare the WO<sub>3</sub> nanorods. After deposition, the WO<sub>3</sub> nanorod thin films were treated with O<sub>2</sub> plasma at different treatment power from 100 – 200 W. The physical structure of as-deposition and treated WO<sub>3</sub> nanorod thin films was investigated crystal structure and morphology by grazing incident X-ray diffraction (GIXRD), field-emission scanning electron microscope (FE-SEM), and high-resolution transmission electron microscope (HRTEM). The result indicated that the WO<sub>3</sub> nanorod structure transformed to the monoclinic polycrystalline phase. FE-SEM and HRTEM observed slight changes in the shape of the WO<sub>3</sub> nanorods. The H<sub>2</sub>S sensing properties were measured at 10 ppm at 150 – 350°C operating temperatures. At an operating temperature of 200 °C, the response to H<sub>2</sub>S of O<sub>2</sub> plasma treated WO<sub>3</sub> nanorods is increased by a factor of 5 – 15, and the maximum response to H<sub>2</sub>S is 15. The results showed that the O<sub>2</sub> plasma treatment process improved the sensing response of the WO<sub>3</sub> nanorods.



**Keywords:** WO<sub>3</sub> nanorods; Oxygen plasma; H<sub>2</sub>S sensor; GLAD technique

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## Introduction

Tungsten Oxide (WO<sub>3</sub>) is an n-type semiconductor and has an energy gap of approximately 2.60 – 3.60 eV, which can be applied in many technologies such as electrochromic, photocatalytic, photovoltaic, and gas-sensor technology. For the gas-sensor technology, the semiconductor gas sensing of WO<sub>3</sub> has been widely studied since it has high sensitivity, high stability, and fast response to many gases such as ethanol, NO<sub>2</sub>, H<sub>2</sub>, and H<sub>2</sub>S [1 – 3]. Moreover, WO<sub>3</sub> can be fabricated into many nanostructure patterns, such as nanotubes, nanowires, nanorods, etc. The WO<sub>3</sub> nanostructured gas sensors were effective for high sensitivity and fast response due to their large surface area and optimum crystallinity [4 – 6]. Increasing the surface area and crystallization by plasma treatment is an interesting technique [7]. Several processes were used to synthesize

WO<sub>3</sub> nanostructures, such as physical vapor deposition (PVD), chemical vapor deposition (CVD), thermal oxidization, flame spray pyrolysis, and hydrothermal [8 – 10]. However, there are many processes to prepare WO<sub>3</sub> nanostructures that cannot control their sizes, distributions, formation, and reproducibility. Therefore, a physical vapor deposition (PVD) method with the GLAD technique is most suitable for synthesizing WO<sub>3</sub> nanostructures [11, 12]. The GLAD technique is an oblique angle deposition (OAD) method with substrate rotation to synthesize a columnar nanorod structure. The porosity and shape of the nanorod structure can be controlled by the self-shadowing effect and surface diffusion of the initial nucleation island during the deposition process [13 – 15].

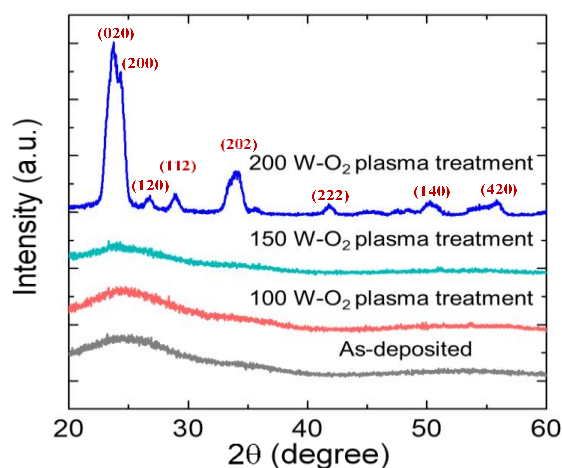
## Materials and Methods

The  $\text{WO}_3$  nanorods were synthesized, and the  $\text{O}_2$  plasma treatment was processed by a reactive DC magnetron sputtering system with the GLAD technique, as shown in the symmetric diagram in Fig. 1. The silicon wafers (100) were used as substrates for physical property analysis, and an ultrasonic washer cleaned them in acetone. Then, the  $\text{SiO}_2/\text{Si}$  with interdigitated gold electrodes were used as substrates for  $\text{H}_2\text{S}$  gas sensor testing. For the sputtering process, the sputtering target was a metallic tungsten disc, and the substrate was 7 cm from the target. Then, the substrate was installed at an angle with the vapor incident flux equal to  $85^\circ$  and the substrate rotation speed fixed at 5 rpm. The  $\text{WO}_3$  nanorods are prepared under the atmosphere with argon (Ar) and  $\text{O}_2$  flow rates of 12 and 48 sccm, respectively. The base pressure and operated pressure are  $6 \times 10^{-6}$  and  $5 \times 10^{-3}$  mbar, respectively. The deposition time was fixed at 90 min. After growth, the  $\text{WO}_3$  nanorod samples were treated with the  $\text{O}_2$  plasma at different power from 100 – 200 W for 10 min. The crystal structures of the  $\text{WO}_3$  nanorods were analyzed by the GIXRD technique, obtained from a grazing incident X-ray diffractometer operating with an incident angle of  $20^\circ - 60^\circ$ . The morphology of the  $\text{WO}_3$  nanorods was analyzed by a field-emission scanning electron microscope (FE-SEM) and a high-resolution transmission electron microscope (HRTEM). For  $\text{H}_2\text{S}$  gas sensor testing, the tungsten oxide nanorods were deposited on interdigitated electrodes and mounted in a gas-testing chamber. The purified air and  $\text{H}_2\text{S}$  gas were mixed with the concentration of  $\text{H}_2\text{S}$  gas at 10 ppm and made to flow through the testing chamber. Then, the  $\text{H}_2\text{S}$  gas sensor was tested with an

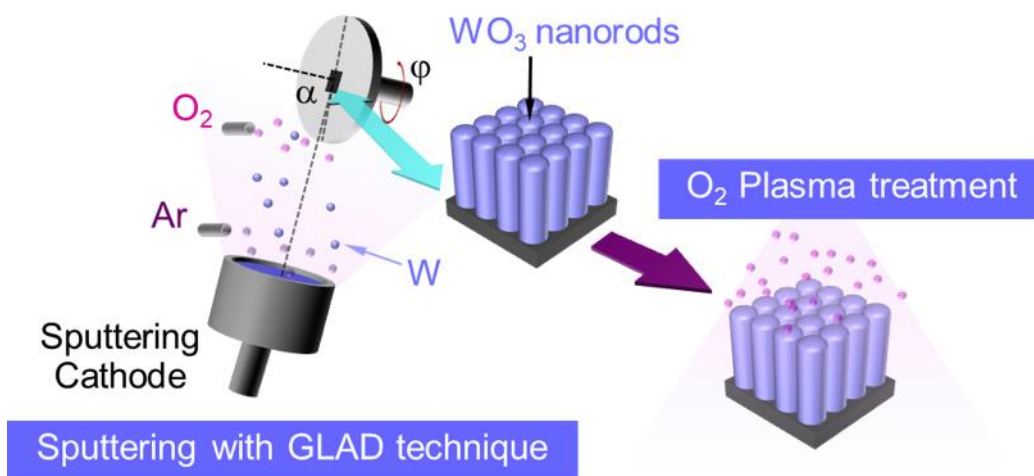
operating temperature varied from 150 – 350  $^\circ\text{C}$  and heated by an external heater.

## Results and Discussions

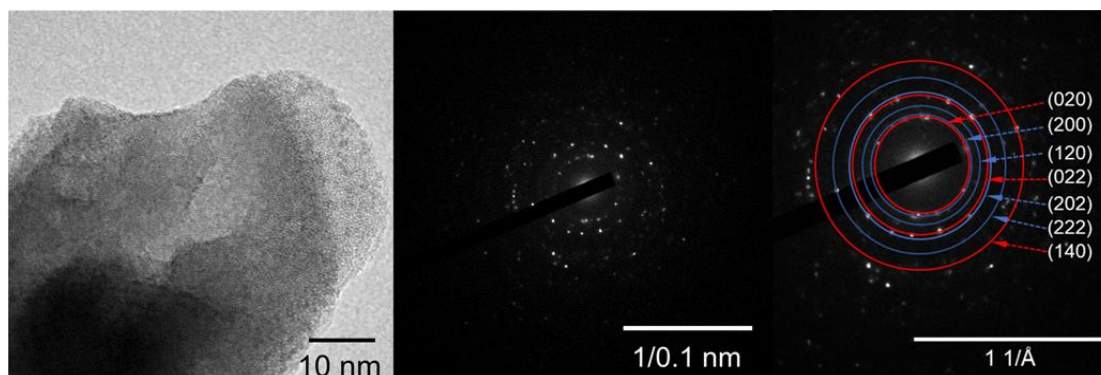
The GIXRD pattern of as-deposited and  $\text{O}_2$  plasma-treated  $\text{WO}_3$  nanorods is shown in Fig. 2. It can be found that the as-deposited and plasma-treated  $\text{WO}_3$  nanorods at 100 – 150 W exhibit an amorphous structure. When the  $\text{O}_2$  plasma treatment is increased to 200 W, the polycrystalline structure of the  $\text{WO}_3$  monoclinic phase can be observed. This might be because the plasma treatment could enhance the internal heating of the sample during the treatment process. Generally, the crystalline transition from amorphous to polycrystalline  $\text{WO}_3$  occurs when the samples are annealed at 300 – 400  $^\circ\text{C}$  [15].



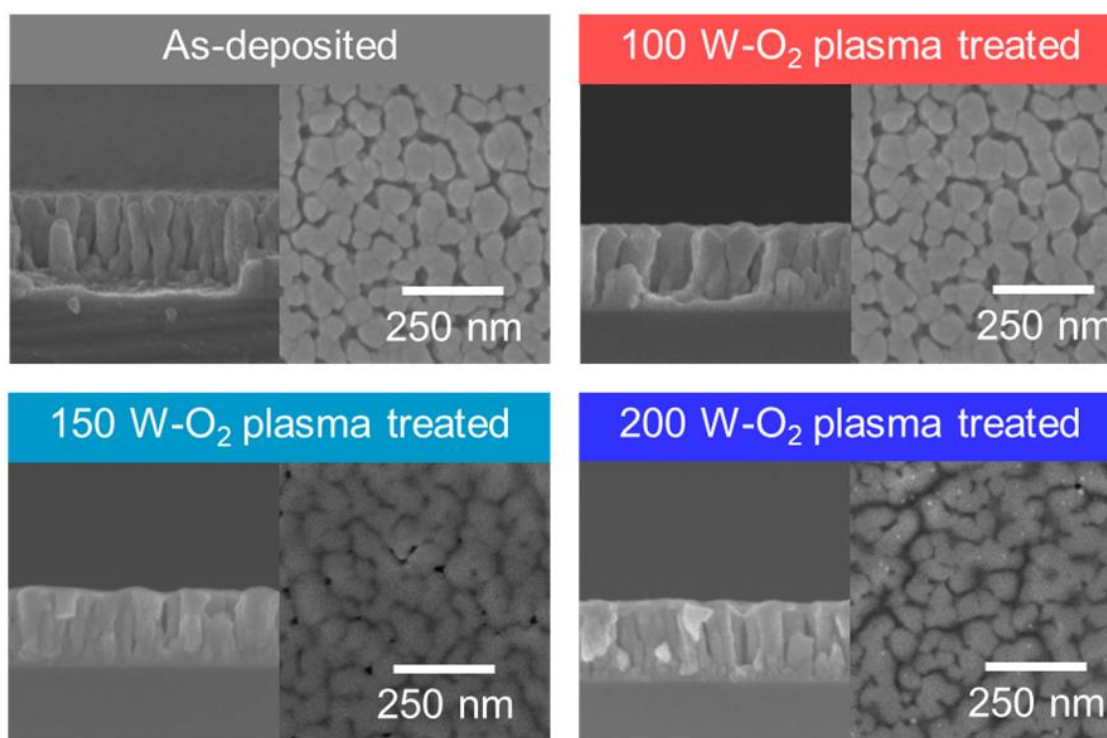
**Fig. 2** GIXRD patterns of as-deposited and oxygen plasma-treated  $\text{WO}_3$  nanorods.



**Fig. 1** Schematic diagram of the GLAD technique and  $\text{O}_2$  plasma treatment process.



**Fig. 3** HRTEM image of 200 W-O<sub>2</sub> plasma-treated WO<sub>3</sub> nanorods.

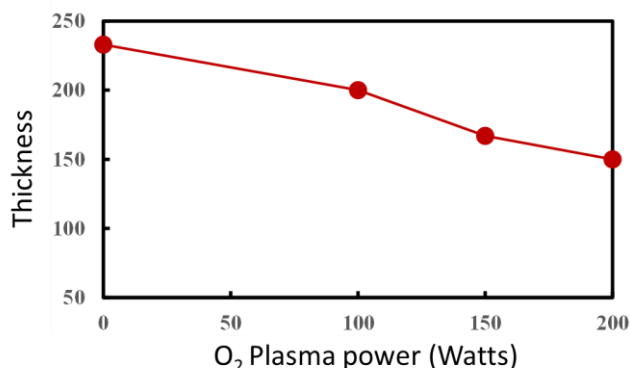


**Fig. 4** FE-SEM images of as-deposited and oxygen plasma-treated WO<sub>3</sub> nanorods.

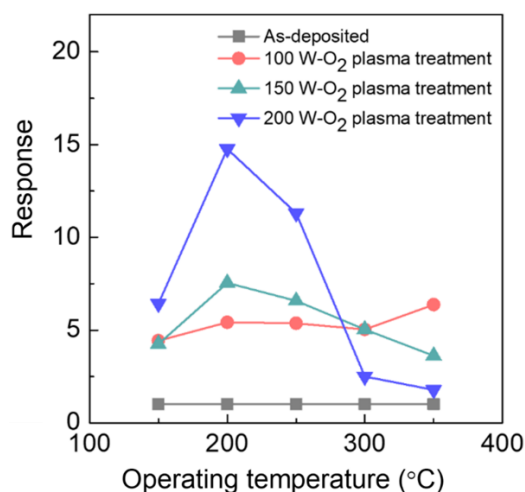
Fig. 3 shows the typical HRTEM and selected area diffraction electron (SADE) images from the tip of the O<sub>2</sub> plasma-treated WO<sub>3</sub> nanorods at 200 W. The crystalline structure of the WO<sub>3</sub> nanorods was clearly observed in the HRTEM image. In addition, the SADE pattern indicates that the plasma-treated WO<sub>3</sub> nanorods at 200 W were polycrystalline, which corresponds to the GIXRD result in the previous section.

The FE-SEM images of as-deposited and O<sub>2</sub> plasma-treated WO<sub>3</sub> nanorods are shown in Fig. 4. The results indicate that all the thin films of tungsten oxide nanorods show columnar nanorod structures. The average diameter of WO<sub>3</sub> nanorods is estimated to be 60 nm, and the WO<sub>3</sub> nanorods thickness of as-deposited and O<sub>2</sub> plasma-treated WO<sub>3</sub> nanorods at 100, 150, and 200 W were measured to be approximately 233, 200, 267 and 150 nm, respectively, as

shown in Fig. 5. However, the surface morphology of the plasma-treated  $\text{WO}_3$  nanorods at high power showed a smooth surface. In addition, the thickness of the  $\text{WO}_3$  nanorod layer decreased with the increase in  $\text{O}_2$  plasma treatment power. The decrease in the smooth surface and etching thickness layer might be due to the  $\text{O}_2$  interaction with the nanorod surface, which indicates that the energy of the  $\text{O}_2$  plasma plays a major role in morphology control.



**Fig. 5**  $\text{WO}_3$  nanorods thickness of as-deposited and  $\text{O}_2$  plasma-treated at 100, 150 and 200 W.



**Fig. 6**  $\text{H}_2\text{S}$  gas response of as-deposited and oxygen plasma-treated  $\text{WO}_3$  nanorods.

As shown in Fig. 6, as-deposited and  $\text{O}_2$  plasma-treated  $\text{WO}_3$  nanorods were measured toward 10 ppm- $\text{H}_2\text{S}$  gas at operating temperatures ranging from 150 – 350 °C. The as-deposited  $\text{WO}_3$  nanorods showed an inadequate response. It can be found that the  $\text{H}_2\text{S}$  gas responses of plasma-treated  $\text{WO}_3$  nanorods increase as the operating temperature increases from 150 – 200 °C. The response to  $\text{H}_2\text{S}$  of  $\text{O}_2$

plasma treatment  $\text{WO}_3$  nanorods is increased by about a factor of 5 – 15 depending on temperature and the adsorbed surface oxygen species. In addition, the  $\text{O}_2$  plasma treatment of  $\text{WO}_3$  nanorods exhibits a maximized response to  $\text{H}_2\text{S}$  of 15 at an operating temperature of 200 °C. The experimental results indicate that  $\text{O}_2$  plasma treatment significantly enhances the  $\text{H}_2\text{S}$  sensing performance by a 380% improvement compared to a dense  $\text{WO}_3$  film. This improvement can be attributed to the modifications in the structure and the increased surface area achieved through the  $\text{O}_2$  plasma treatment [17 – 20].

## Conclusion

In conclusion, we present the crystal structure, morphology, and  $\text{H}_2\text{S}$  sensing of as-deposited and oxygen plasma-treated  $\text{WO}_3$  nanorods on Si substrate. The post- $\text{O}_2$  plasma treatment at 200 W promoted the polycrystalline  $\text{WO}_3$  nanorods. The plasma-treated  $\text{WO}_3$  nanorods at high  $\text{O}_2$  plasma power tend to smooth surface and decrease in thickness. In addition, the plasma-treated  $\text{WO}_3$  nanorods at 200 W were revealed to be promising for achieving  $\text{H}_2\text{S}$  sensors with a high response of about 15 at 200 °C optimum operating temperature compared with as-deposited  $\text{WO}_3$  nanorods.

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## References

- [1] Y. Shen, T. Yamazaki, Z. Liu, D. Meng, T. Kikuta, N. Nakatani, Influence of Effective Surface Area on Gas Sensing Properties of  $\text{WO}_3$  Sputtered Thin Films, *Thin Solid Films*. 517 (2009) 2069 – 2072.
- [2] D. Deniz, D.J. Frankel, R.J. Lad, Nanostructured tungsten and tungsten trioxide films prepared by glancing angle deposition, *Thin Solid Films*. 518 (2010) 4095 – 4099.
- [3] M. Horprathuma, K. Limwicheana, A. Wisitsoraat, P. Eiamchai, K. Aiempantakit, P. Limnonthakul, N. Nuntawong, V. Pattantsetakul, A. Tuantranont, P. Chindaudom,  $\text{NO}_2$ -sensing properties of  $\text{WO}_3$



- nanorods prepared by glancing angle DC magnetron sputtering, *Sens. Actuators B Chem.* 176 (2013) 685 – 691.
- [4] Y.B. Li, Y. Bando, D. Golberg, K. Kurashima, WO<sub>3</sub> Nanorods/Nanobelts Synthesized via Physical Vapor Deposition Process, *Chem. Phys. Lett.* 367 (2003) 214 – 218.
- [5] C. Oros, M. Horprathum, A. Wisitsoraat, T. Srichaiyaperk, B. Samransuksamer, S. Limwichean, P. Eiamchai, D. Phokharatkul, N. Nuntawong, C. Chananon-nawathorn, V. Patthanasettakul, A. Klamchuen, J. Kaewkhao, A. Tuantranont, P. Chindaudom, Ultra-sensitive NO<sub>2</sub> sensor based on vertically aligned SnO<sub>2</sub> nanorods deposited by DC reactive magnetron sputtering with glancing angle deposition technique, *Sens. Actuators B Chem.* 223 (2016) 936 – 945.
- [6] H. Du, J. Wang, Y. Sun, P. Yao, X. Li, N. Yu, Investigation of gas sensing properties of SnO<sub>2</sub>/In<sub>2</sub>O<sub>3</sub> composite hetero-nanofibers treated by oxygen plasma, *Sens. Actuators B Chem.* 206 (2015) 753 – 763.
- [7] Y. Hou, A.H. Jayatissa, Enhancement of gas sensor response of nanocrystalline zinc oxide for ammonia by plasma treatment, *Appl. Surf. Sci.* 309 (2014) 46 – 53.
- [8] A. Sharma, M. Tomar, V. Gupta, SnO<sub>2</sub> thin film sensor with enhanced response for NO<sub>2</sub> gas at lower temperatures, *Sens. Actuators B Chem.* 156(2) (2011) 743 – 752.
- [9] H.J. Gwon, H.G. Moon, H.W. Jang, S.J. Yoon, K.S. Yoo, Sensitivity enhancement of nanostructured SnO<sub>2</sub> gas sensors fabricated using the glancing angle deposition method, *J. Nanosci. Nanotechnol.* 13(4) (2013) 2740 – 2744.
- [10] C. Zhang, J. Wang, X. Geng, Tungsten oxide coatings deposited by plasma spray using powder and solution precursor for detection of nitrogen dioxide gas, *J. Alloys Compd.* 668 (2016) 128 – 136.
- [11] M.Z. Ahmada, A. Wisitsoraat, A.S. Zoolfakar, R.A. Kadir, W. Wlodarski, Investigation of RF sputtered tungsten trioxide nanorod thin film gas sensors prepared with a glancing angle deposition method toward reductive and oxidative analytes, *Sens. Actuators B Chem.* 183 (2013) 364 – 371.
- [12] M. Horprathum, T. Srichaiyaperk, B. Samransuksamer, A. Wisitsoraat, P. Eiamchai, S. Limwichean, C. Chananon-nawathorn, K. Aiempnanakit, N. Nuntawong, V. Patthanasettakul, C. Oros, S. Porntheeraphat, P. Songsiriritthigul, H. Nakajima, A. Tuantranont, P. Chindaudom, Ultrasensitive Hydrogen Sensor Based on Pt-Decorated WO<sub>3</sub> Nanorods Prepared by Glancing-Angle dc Magnetron Sputtering, *ACS Appl. Mater. Interfaces.* 6 (2014) 22051 – 22060.
- [13] H.W. Kim, S.W. Choi, A. Katoch, S.S. Kim, Enhanced sensing performances of networked SnO<sub>2</sub> nanowires by surface modification with atmospheric pressure Ar–O<sub>2</sub> plasma, *Sens. Actuators B Chem.* 177 (2013) 654 – 658.
- [14] Y.H. Liang, W. Jun, Z. Biao, Z.B. Wen, L.L. Qin, Z.Y. Wu, Z. Cheng, X.X. Ling, Fabrication of single crystalline WO<sub>3</sub> nano-belts based photoelectric gas sensor for detection of high concentration ethanol gas at room temperature, *Sens. Actuator A Phys.* 303 (2020) 11 – 18.
- [15] K. Khojier, S. Zolghadr, F. Teimoori, S. Goudarzi, Fabrication and characterization of porous WO<sub>3</sub> thin film as a high accuracy cyclohexene sensor, *Mater. Sci. Semicond. Process.* 118 (2020) 115 – 220.
- [16] M. O'Brien, K. Lee, R. Morrish, N. C. Berner, N. McEvoy, C. A. Wolden, G. S. Duesberg, Plasma assisted synthesis of WS<sub>2</sub> for gas sensing applications, *Chem. Phys. Lett.* 615 (2014) 6 – 10.
- [17] Y. Zhang, J. Li, G. An, X. He, Highly porous SnO<sub>2</sub> fibers by electrospinning and oxygen plasma etching and its ethanol-sensing properties, *Sens. Actuators B Chem.* 44 (2010) 43 – 48.
- [18] B. Urasinska-Wojcik, T.A. Vincent, J.W. Gardner, H<sub>2</sub>S sensing properties of WO<sub>3</sub> based gas sensor, *Procedia Engineering.* 168 (2016) 255 – 258.
- [19] M. Takácsa and A.E. Pap, Gas sensitivity of sol-gel prepared mesoporous WO<sub>3</sub> thin film, *Procedia Engineering.* 168 (2016) 289 – 292.
- [20] T. Shujah, M. Ikram, A. R. Butt, M.K. Shahzad, K. Rashid, Q. Zafar, S. Ali, H<sub>2</sub>S Gas Sensor Based on WO<sub>3</sub> Nanostructures Synthesized via Aerosol Assisted Chemical Vapor Deposition Technique, *nanoscience and Nanotechnology Letters.* 11 (2019) 1 – 10.